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LASER-INDUCED FLUORESCENCE LINE NARROWING OF EU3+ IN LITHIUM BORATE GLASS

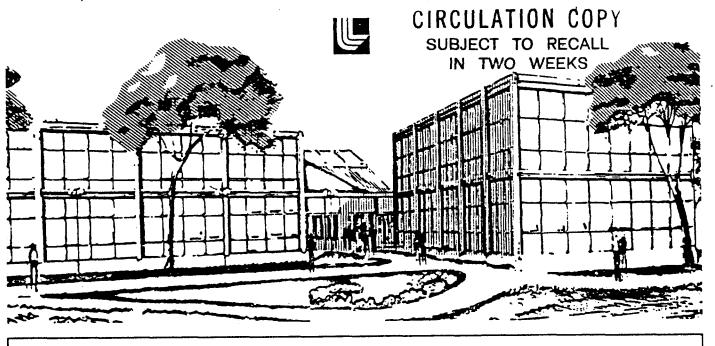
J. Hegarty and W. M. Yen
Department of Physics, University of Wisconsin-Madison
Madison, Wisconsin 53706 USA

M. J. Weber Lawrence Livermore Laboratories, University of California Livermore, California 94550 USA

D. H. Blackburn Inorganic Glass Section, National Bureau of Standards Washington, D.C. 20234 USA

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Laser-induced fluorescence line narrowing of Eu³⁺ in Lithium Borate glass*

J. Hegarty and W. M. Yen

Department of Physics, University of Wisconsin-Madison

Madison, Wisconsin 53706 USA

M. J. Weber**

Lawrence Livermore Laboratories, University of California Livermore, California 94550 USA

D. H. Blackburn

Inorganic Glass Section, National Bureau of Standards
Washington, D.C. 20234 USA

Abstract

Laser-induced fluorescence line narrowing reveals structural changes in Eu³⁺ doped borate glasses as a function of Li₂O network modifier concentration. These changes are discussed in terms of different coordination and bonding and possible phase separation.

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Laser-induced fluorescence line narrowing has recently been used to investigate the local environment and interactions of paramagnetic ions in glass. Since the resulting spectra are sensitive to structural modifications, this technique has been applied to simple alkali borate glasses (1,2) where the relative numbers of network-forming BO₃ triangles and BO₄ tetrahedra (3) and the existence of subliquidus immiscibility(4,5) are dependent upon the mole fraction of alkali oxide.

Paramagnetic ions generally enter a glass as interstitial network-modifying cations, not as substitutional ions. Because of differences in the bonding to near neighbours the local fields at individual ion sites vary, resulting in large inhomogeneous linewidths in absorption and emission. In the past, analysis of structure arising from changes in borate glass composition was hampered by the simultaneous observation of many sites. This difficulty can be overcome by the use of a narrowband laser source(6). In our studies trivalent europium is used as the probe ion because it has a relatively simple energy level scheme.

Previous investigations(1,2) of lithium borate glasses revealed large differences in the FLN spectra of Eu³⁺ for samples containing 10 and 40 mole percent of Li₂O. Whereas the spectra from the two samples were similar for laser excitation on the low energy side of an inhomogeneously broadened band, at high excitation energies additional lines appeared in the spectra from the 10 mole % Li₂O glass. Among the possible causes of this behaviour is phase separation into, say, a lithia-rich and a borate-rich phase. To examine the possibility further, we have investigated glasses with smaller Li₂O contents approaching the case of simple B₂O₃.

Details of the glass preparation procedure and of the experimental techniques used are outlined in Ref. 1. In the case of glass preparation,

attempts to make clear $\mathrm{Eu_2O_3}$ -doped $\mathrm{B_2O_3}$ glasses failed; glasses with concentrations as low as 0.01 mol.% $\mathrm{Eu_2O_3}$ and 99.99 mol.% $\mathrm{B_2O_3}$, though clear at elevated temperatures in the melt, would change color on cooling indicating a type of phase separation. $\mathrm{Li_2O}$ was added until a clear glass was formed with 0.1 mol.% $\mathrm{Eu_2O_3}$ and the remainder $\mathrm{B_2O_3}$. The lowest concentration of $\mathrm{Li_2O}$ that would form a clear glass on cooling was 3 mol.%.

The laser-induced FLN spectra from the 5D_o state of Eu $^{3+}$ to the 7F_1 and 7F_2 manifolds are shown in Fig. 1 for 3 and 40 mol.% Li $_2$ O borate samples. The laser excitation energy (cm $^{-1}$) is given at the left of each spectrum (the spectra are not normalized to constant excitation energy). The fluorescence spectra in Fig. 1 show some line narrowing, but the widths are still greater than either the homogeneous or instrumentation widths because the spectra were recorded with nonresonant excitation.

The change in line positions in the FLN spectra with excitation energy arise from differences in the local fields. As first reported by Motegi and Shionoya, 7 the variation in crystal-field splitting exhibits a monotonic change with $^7F_0^{-5}D_0$ excitation energy. The low-energy excitation portions of the FLN spectra for the two samples are very similar suggesting that they arise from Eu $^{3+}$ ions having equal coordination and bonding. The high-energy excitation portions, however, are very different. In addition, extra lines appear in the 3% Li $_2$ O sample which are identical to those observed previously in a 10% Li $_2$ O sample 1 and which are attributed to Eu $^{3+}$ ions in sites having a different local environment. These lines are labeled 3, 4 and 7 in Fig. 1.

The fields at Eu³⁺ sites are expected to change as Li₂O content is

increased because boron with three coordinating oxygens is replaced by boron with four coordinating oxygens and by Li⁺ network modifying cations. An increase in the Eu-O covalency reduces the Racah parameters which, in turn, lowers the ${}^{5}D^{-7}F$ energy separation. A similar effect has been observed for Nd^{3+} in $x Li_2^{0}$ (1-x) B_2^{0} glasses(8) where the 4I - 4F separation decreases with increasing $\text{Li}_2^{\circ}0$ content for $0 < x \le 0.45$. This is already apparent from the shift of the peak at the strong ${}^4F_{3/2} \rightarrow {}^4I_{11/2}$ fluorescence band at 300 K. Therefore, in the case of Eu3+ the low energy side of the inhomogeneous $^{7}F_{0}^{-5}D_{0}$ transition corresponds to sites with stronger Eu-O interaction and covalency. Conversely, one would expect that the high-excitation-energy spectra should arise from sites where BO3 triangles are prevalent. At 40% $\mathrm{Li}_2\mathrm{O}$ concentration, where the maximum number of BO_3 triangles are converted into BO_4 tetrahedra, one would expect to see evidence of Eu^{3+} in sites of both coordinations. The spectra, however, vary monotonically with pump energy and do not exhibit any dramatic change at high pump energies. In addition, the lifetimes of all of the luminescence lines at any one pump wavelength are identical in this sample indicating that one type of coordination is predominant. In the 3 and 10% samples, the additional lines appearing at high pump energies have lifetimes different from those of the regular lines so that two types of sites are being simultaneously pumped.

Metastable immiscibility on a scale of 5-500 nm is known to occur in the lithium borate system. 4,5 Lithium-rich and 20 3-rich phases are expected to be present in the 10% 20 0 sample since the maximum of the coexistence curve occurs at this concentration. If impurities such as 20 4 enter one or both phases, the excitation energies may be sufficiently different to permit

relative probing of local structure in the individual phases. If two phases are present in the 10% Li₂O sample, the low-energy FLN spectra suggest that one phase is similar to that of the 40% Li₂O sample. On the basis of the nephelauxetic effect, such a phase is likely to be the Li₂O-rich phase, where the concentration of BO₄ tetrehdra may be of the same order as in the 40% sample. Further support for this theory is found in samples of lower Li₂O concentration. The 3% Li₂O sample behaves identically to the 10% sample but it is noteworthy that for a Li₂O concentration of 3% the maximum Eu doping possible is only 0.2%. Consequently, at such concentrations the relative numbers of Eu ions and Li ions present are roughly similar to those of the 10% sample.

Europium ions cannot be introduced into simple B_2O_3 glass in the absence of Li_2O . The network of $(BO_3)^3$ units in glassy B_2O_3 is tightly bonded by bridging oxygens. Trivalent Eu ions cannot enter this network substitutionally (because of size), nor, apparently, as a network modifier. If, however, a network modifier such as Li^+ is present, it disrupts the network and non-bridging oxygens occur. In the vicinity of the looser structure, Eu^{3+} may be incorporated as modifier ions.

Additional spectral lines could also arise from clustering of Eu^{3+} ions on a Eu^{3+} rich phase. If ion-ion interactions are strong, they can alter the energy levels from those for isolated Eu^{3+} ions.

References

- 1. M. J. Weber, J. Hegarty, D. H. Blackburn in Boron in Glass and Glass Ceramics (Plenum Press, New York 1978), in press.
- 2. J. Hegarty, W. M. Yen and M. J. Weber, Phys. Rev. B (1978) in press.
- 3. W. L. Konijnendik, Philips Res. Repts. Suppl. No. 1 (1975).
- 4. E. M. Levin, Phys. Chem. Glasses <u>7</u>, 90 (1966).
- 5. R. R. Shaw and D. R. Uhlmann, J. Amer. Ceram. Soc. <u>51</u>, 377 (1968).
- L. A. Riseberg, Phys. Rev. Lett. <u>28</u>, 789 (1972); Phys. Rev. A7, 671 (1973).
- 7. N. Motegi and S. Shionoya, J. Luminescence 8, 1 (1973).
- 8. M. J. Weber, unpublished.

Figure Captions

Fig. 1. Comparison of the 5D_0 - 7F_1 , 7F_2 fluorescence spectra of Eu³⁺ in 3 Li₂O · 97 B₂O₃ and 40 Li₂O · 60 B₂O₃ glasses at 1.7K as a function of 7F_0 - 5D_0 excitation frequency (cm⁻¹).

